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Intercalation and formation of complexes in the system of lead(II) iodide–ammonia

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Abstract

Interaction between lead(II) iodide and ammonia was studied with the help of an X-ray in situ analysis, DTA-TG analysis, DSC measurements and IR spectroscopy. A two-stage mechanism of the reaction was defined. At the first stage of the reaction two phases with trigonal symmetry and a phase with monoclinic symmetry are developed. At the second stage of the reaction the structure changes lead to formation of a compound with orthorhombic symmetry. The results were discussed along with the data of thermal analysis and IR spectroscopy. The value of enthalpy of formation for the compound $PbI_2(NH_3)_4$ was determined. \odot 2003 Elsevier Inc. All rights reserved.

Keywords: Lead(II) iodide; Ammonia; Intercalation; Complex

1. Introduction

Intercalation of various electron-donating molecules into interlayer of $PbI₂$ crystals having a layered structure has been studied widely in recent decades [\[1–16\].](#page-5-0) In particular, thermodynamics of forming intercalated compounds of $PbI₂$ [\[9,17,18\]](#page-5-0), photochemical reactions of intercalated compounds and photolysis of $PbI₂$ sensitized by molecules of intercalates [\[17,19–21\],](#page-5-0) as well as properties of intercalated phases of PbI₂ [\[8,18,22,23\]](#page-5-0) were studied. There was shown the possibility of using PbI_2 for quantitative determination of amines in non-reducing media [\[19\].](#page-5-0) In this connection a study of PbI_2 –intercalate systems, in which formation of several intercalated phases occurs, is of special interest. PbI_2 -2-aminoethanol [\[8,20\],](#page-5-0) PbI_2 -pyridine [\[1,9,22\]](#page-5-0), $PbI₂$ –hexahydropyridine [\[1,14\]](#page-5-0) and $PbI₂$ –ammonia [\[12,20,21,23\]](#page-5-0) are examples of such systems.

An X-ray analysis in situ was carried out for the system of PbI_2-NH_3 [\[24–26\]](#page-5-0) and $PbI_2-C_5H_{11}N$ [\[14\]](#page-5-0). Changes of the unit-cell parameters of $PbI₂$ crystal matrix [\[24\]](#page-5-0) and development of a metastable phase were

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determined [\[25,26\]](#page-6-0). Kinetics of the changes in diffraction patterns on incorporation of hexahydropyridine molecules into textured films of $PbI₂$ were studied [\[14\].](#page-5-0)

In this paper, we speak about our study of consequent changes in crystal structure of lead iodide and in properties of intercalate during incorporating NH₃ molecules. Methods of an X-ray structure analysis in situ, DTA-TG analysis, differential scanning calorimetry (DSC) and infrared spectroscopy were used.

2. Experimental

The purity of PbI_2 powder was checked by an X-ray analysis and IR spectroscopy. The purity of gaseous NH3 was checked by the IR spectroscopy method. An X-ray study was carried out with the help of two powder X-ray diffractometers using $CuK\alpha$ radiation. The lattice parameters and their standard deviations were calculated using a dichotomy method [\[27\]](#page-6-0) after least-squares refinement of peak positions. The profiles of the diffraction lines were fitted with a Pseudo-Voigt function. To index the powder diffraction patterns a DICVOL93 program (D. Louer and A. Boultif), to calculate the theoretical X-ray powder diffraction patterns a LAZYPULVERIX program (K. Yvon) and

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to fit the profiles of diffraction lines a DIFFRAC/AT program (SOCABIM) there were used.

A trigonal system and a *P3ml* space group for a pure PbI₂ powder with the hexagonal unit-cell parameters of $a = 4.5569(4)$ Å and $c = 6.9803(12)$ Å were revealed. The lattice parameters were in a good concordance with the data of the work [\[28\]](#page-6-0) and corresponded to 2H polytype of PbI₂. After deammoniation a restoration of the $PbI₂$ structure was observed.

To perform an X-ray analysis in situ in course of interaction between PbI_2 and NH_3 the sample of PbI_2 powder was placed into a special vacuum chamber equipped with a valve, an inlet for gaseous ammonia supply and a gas-pressure-measuring system. At a lower temperature a decrease in the reaction rate occurs which made it possible to record the diffraction patterns at the intermediate stages of the reaction. Our X-ray diffraction experiments were performed at a temperature of 285 K. The previously evacuated chamber was filled with gaseous ammonia up to a pressure of about 760 Torr.

Calorimetric measurements were carried out by the DSC with the help of a diathermal-shell calorimeter. A sample was prepared in a small glass ampoule. The previously evacuated ampoule with $PbI₂$ powder was filled with gaseous ammonia up to a pressure of 760 Torr and maintained under that condition for 96 h. To avoid decomposition, the ampoule was cooled to the liquid nitrogen temperature and was immediately sealed. The exposure time of a PbI_2 sample in ammonia medium was chosen to be 96 h, since at the longer exposure times no changes in the diffraction patterns were observed. The heating rate of the sample was equal to 5 K min^{-1} during the DSC measurements.

Besides, the methods of differential thermal analysis (DTA) and thermogravimetric analysis (TG) in argon atmosphere employing a Paulic–Paulic–Erday thermoanalyser were used. The heating rate was equal to 5 K min⁻¹. A double-beam "Specord IR-75" spectrophotometer was used in an infrared study. $PbI₂$ powder of was put on a KBr plate. Then the sample was placed into a gas chamber equipped with an inlet for gaseous ammonia supply and a gas-pressure-measuring system.

3. Results and discussion

[Fig. 1A](#page-2-0) shows a diffraction pattern of a pure $PbI₂$ sample. The several diffraction patterns that were obtained during an X-ray analysis in situ in the course of interaction between PbI_2 and ammonia at a temperature of 285 K are presented in [Figs. 1B–E](#page-2-0). Two stages of the reaction can be found. A decrease in pristine PbI₂ content, formation of the intercalated phases I and II and development of the phase III [\(Figs. 1B and C\)](#page-2-0) were revealed at the first stage of the reaction. A decrease in peak intensity of pristine $PbI₂$,

development of the double peak at $2\Theta_1^\circ = 11.60^\circ$ and $2\theta_2^{\circ} = 11.81^{\circ}$ and an increase in peak intensity of the phase III lead to this conclusion.

The appearance of diffraction lines at lower angles is induced by the enlarged size of the unit cell in c -axis, which is a typical manifestation of incorporation of guest molecules into a host matrix [\[2,6\]](#page-5-0). Theoretical diffraction patterns for PbI_2 with the values $c_1 = 7.49$ A (for $2\Theta_1^{\circ}$) and $c_2 = 7.62 \text{ Å}$ (for $2\Theta_2^{\circ}$) in $P\overline{3}ml$ space group were calculated and compared with the diffraction patterns obtained experimentally. A new parameter of the unit cell a_1 was selected using the most intensive lines from the (001), (011), (102), (012), (110), (103) and (201) planes. The value a_1 was found to be the same for the both values c_1 and c_2 , which was equal to 4.92 Å. It exceeded the initial value for PbI_2 lattice by 0.38 Å.

Thus, at the first stage of the reaction the two kinds of the unit cells with enlarged hexagonal parameters $a_1 = 4.92 \text{ Å}, c_1 = 7.62 \text{ Å}$ and $a_1 = 4.92 \text{ Å}, c_2 = 7.49 \text{ Å}$ were formed—phases I and II, respectively. The enlarged size of the unit cell in *a*-axis by $0.2 - 0.4$ A together with the enlarged size in c-axis by $5.8-14.1 \text{ Å}$ was also found [\[14\]](#page-5-0) during hydrazine-intercalation of thin evaporated films and powdered single crystals of $PbI₂$ at a pressure of about 20 Torr.

[Table 1](#page-3-0) shows the X-ray diffraction data with assigned Miller indices for the phase III. Stoichiometry of the phases was determined by DTA–TG analysis and corresponded to the compound of $PbI_2(NH_3)_{2.47}$. The lattice parameters for the phase III were calculated: $a = 10.238(11)$ A, $b = 12.061(8)$ A, $c = 9.587(8)$ A, $\beta =$ 107.73(7)° and $V = 1127.54 \text{ Å}^3$. The lattice symmetry of the phase III was found to be monoclinic.

A decrease in the phase III content and formation of the phase IV ([Figs. 1D–E](#page-2-0)) were found at the second stage of the reaction. A decrease in the peak intensity of the phase III and an increase in the peak intensity of the phase IV supported the above conclusion. It was clear that the phase IV formed at this stage was not an intercalation compound but formation of the complex occurred. Stoichiometry of the phase IV was determined by DTA–TG analysis and corresponded to the compound of $PbI_2(NH_3)_4$.

[Table 2](#page-3-0) shows the X-ray diffraction data with assigned Miller indices for the phase IV. The lattice parameters for the phase IV were calculated: $a = 19.894(23)$ Å, $b = 9.345(27)$ Å, $c = 7.021(10)$ Å and $V = 1305.35 \text{ Å}^3$. The lattice symmetry of the phase IV was found to be orthorhombic.

The thermal analysis data confirmed the existence of several compounds in the system of PbI_2-NH_3 . The results of thermogravimetric study of the phase IV are shown in [Fig. 2B and C.](#page-3-0) The weight loss was consistent with the formulation of the starting composition, i.e. the phase IV, of $PbI_2(NH_3)_4$. Decomposition began at the temperatures $T < T_0$ which was

Fig. 1. X-ray diffraction patterns of PbI₂ powder sample at a temperature of 285 K before and after the beginning of the reaction between PbI₂ and NH₃: (A) pristine PbI₂; (B) 14 h, (C) 24 h, (D) 48 h; and (E) 72–96 h after the beginning of the reaction.

determined by instability of $PbI_2(NH_3)_4$ without ammonia medium. Fractions in the curve B [\(Fig. 2\)](#page-3-0) occurred at temperatures of $T_2 = 359 \text{ K}$, $T_3 = 393 \text{ K}$ and $T_4 = 423$ K. At the temperatures of T_2 and T_4 the rate of weight loss of the sample reached maximum values ([Fig. 2C\)](#page-3-0). So the point T_2 corresponded to a maximum temperature of decomposition of the phase III, and the point T_4 to a maximum temperature of

decomposition of the phases II and I. The weight loss at the point T_2 was consistent with the formulation of the phase III of $PbI_2(NH_3)_{2.47}$; the weight loss at the point T_4 to with the formulation of the phases II and I of $PbI_2(NH_3)_{1.3}$. Probably the difference between the phases II and I was due to the orientation of the $NH₃$ molecules in the gallery space of the layered structure of PbI₂. The phase III differed from phases I and II in an

Table 1 X-ray data for the phase III —PbI₂(NH₃)_{2.47}

h k l	$d_{\text{OBS}}(\AA)$	$d_{\text{CAL}}(\AA)$	$d_{\text{OBS}} - d_{\text{CAL}}(A)$	I/I_0 (%)	$2\Theta_{\rm OBS}$
110	7.557	7.58314	-0.02559	78.1	11.70
020	6.021	6.03066	-0.00941	15.7	14.70
$20\bar{1}$	4.979	4.97658	0.00239	58.8	17.80
012	4.271	4.26993	0.00126	37.2	20.78
121	4.207	4.19422	0.01292	34.3	21.10
$20\bar{2}$	3.995	3.99434	0.00142	62.7	22.23
201	3.834	3.84223	-0.00812	84.3	23.18
$22\bar{1}$		3.83840	-0.00429		
031	3.690	3.67957	0.01022	82.3	24.10
211	3.660	3.66096	-0.00109	76.5	24.30
$13\bar{1}$	3.580	3.59128	-0.01119	23.5	24.85
$20\bar{3}$	3.025	3.02942	-0.00391	78.1	29.50
032		3.01731	0.00819		
$32\bar{1}$	2.966	2.96678	-0.00023	77.5	30.10
$21\bar{3}$	2.938	2.93816	-0.00021	40.2	30.40
231	2.778	2.77773	-0.00002	13.7	32.20
$32\overline{2}$		2.78344	-0.00573		
141	2.677	2.67899	-0.00228	58.8	33.45
$24\bar{1}$	2.581	2.57888	0.00205	50	34.73
$11\bar{4}$	2.342	2.34453	-0.00224	100.0	38.40
142		2.34304	-0.00076		
150		2.34168	0.00061		
004	2.285	2.28281	0.00230	87.5	39.40
203		2.28787	-0.00276		
322	2.176	2.17620	0.00000	46.9	41.46
$15\bar{2}$	2.148	2.14838	0.00011	50.0	42.02
104	2.086	2.08618	-0.00058	21.9	43.35

Table 2 X-ray data for the phase IV —PbI₂(NH₃)₄ hkl d_{OBS} (\AA) d_{CAL} (\AA) d_{OBS} - d_{CAL} (\AA) I/I₀ (%) 0 1 1 5.629 5.61348 0.01574 42.8 15.73

 $2\Theta_{\rm OBS}$

increased ammonia content and another type of cell symmetry.

Endothermic effect was manifested during the outflow of ammonia from a sample, and on the DTA traces the maximums were fixed at the temperatures T_2 and T_4 (Fig. 2D). The temperature region T_0-T_2 corresponds to the transition from the phase IV to the phase III, and the region $T_2 - T_4$ to the transition from the phase III to the phases II and I. In the temperature region $T_4 - T_5$ the transition to a pristine $PbI₂$ structure occurred.

Curve A shown in Fig. 2 was obtained during a differential scanning calorimetry (DSC) study of $PbI_2(NH_3)_4$ sample. In the curve there was a single endothermic peak at the temperature of $T_4 = 423$ K which indicated an deammoniation process in the sample. An absence of any peak at lower temperatures showed that ammonia molecules persisted in the sample at temperatures $T < T_3$. Evidently, at those temperatures the ammonia concentration in a sealed ampoule with the sample was higher than the threshold concentration of $NH₃$ at the beginning of decomposition. At a higher temperature an increase in the threshold concentration occurred. At temperatures $T>T_3$ the threshold concentration exceeded the ammonia concentration in the ampoule, and deammoniation process became possible. The intercalation threshold in PbI_2 for various amines

Fig. 2. Data of thermal analysis for the $PbI_2(NH_3)_4$ samples: (A) differential scanning calorimetric curve; (B) thermogravimetric curve; (C) derived from the slope of the curve B; and (D) DTA curve.

and the dependence of threshold on temperature were studied earlier [\[10,17\]](#page-5-0).

An enthalpy (ΔH) of formation of PbI₂(NH₃)₄ was determined from the DSC data, and was equal to 20.8 kJ mol^{-1} . This value was close to that of other intercalated PbI_2 compounds [\[9,18\]](#page-5-0).

To explore the changes of vibrational spectra of incorporated NH_3 molecules in NH_3 –PbI₂ compounds, an IR spectroscopic study in the range of 400– 4000 cm^{-1} , where the semiconductor matrix PbI₂ is transparent, was carried out. The changes of IR spectra of samples were recorded during deammoniation of $PbI_2(NH_3)_4$ at a temperature of 285 K. During transition from $PbI_2(NH_3)_4$ to the phase III the color of the sample changed from white to pale yellow. Parts of IR spectra of the $PbI_2(NH_3)_4$ and $PbI_2(NH_3)_{2.47}$ compounds during the deammoniation process are presented in Figs. 3A–C. Fig. 3D shows a part of IR spectra of gaseous $NH₃$ within the same region of wavenumbers. Symmetric $v_1(NH)$ and antisymmetric $v_3(NH)$ modes of the N–H bond of gaseous ammonia at 3333 and 3444 cm^{-1} were shifted: (i) to 3320 and 3433 cm⁻¹ for $PbI_2(NH_3)_{2,47}$ and (ii) to 3307 and 3427 cm⁻¹ for $PbI_2(NH_3)_4$, respectively. Observed changes for $PbI_2(NH_3)_{2.47}$ are in good agreement with the data of the work [\[12\]](#page-5-0) for intercalated samples of textured thin

Fig. 3. Parts of infrared spectra: (A) of ammonia in the $PbI_2(NH_3)_4$ sample; (B) of ammonia in the sample during the transition from $PbI_2(NH_3)_4$ to $PbI_2(NH_3)_{2,47}$; (C) of ammonia in the $PbI_2(NH_3)_{2,47}$ sample; and (D) of gaseous ammonia.

films of PbI_2 . These changes of vibrational spectra of incorporated amines were a typical manifestation of the fact of interaction between $PbI₂$ and amine intercalates $[5,7,12,22]$. In the works $[5,7]$ the fact of charge transfer interaction between atom of Pb and free electron pair of nitrogen atom was ascertained for the compounds of $PbI₂$ intercalated by amines.

New modes at 3307 cm^{-1} for $\text{PbI}_2(\text{NH}_3)_{2.47}$ and at 3286 cm^{-1} for PbI₂(NH₃)₄ were found in the range of $v_3(NH)$. The H–N–H torsional mode at 628 cm⁻¹ was slightly shifted to 624 cm^{-1} for $\text{PbI}_2(\text{NH}_3)_{2.47}$, and two new modes at 629 cm^{-1} and 611 cm^{-1} were found for $PbI_2(NH_3)_4$. The mode of NH^{4+} was not observed which was also pointed out in the work [\[12\]](#page-5-0), where IR spectrum in the range of $400-4000 \text{ cm}^{-1}$ was presented.

For $PbI_2(NH_3)_{2.47}$ and $PbI_2(NH_3)_{4}$ the values of the shift in absorption band of symmetric $v_1(NH)$ mode were essentially greater than those of the antisymmetric $v_3(NH)$ mode (Table 3). A similar effect was revealed during transition from gaseous to crystalline ammonia, when hydrogen bonds were formed by free electron pair of nitrogen atom and atom of hydrogen [\[29\].](#page-6-0) During selective adsorption of ammonia molecules on a surface [\[30\]](#page-6-0), when the interaction between surface cations and free electron pair of nitrogen atom of NH_3 mostly occurred, an inverse effect was observed, namely, shift of $v_3(NH)$ mode exceeded shift of $v_1(NH)$ mode. A similar inverse effect was revealed earlier [\[9,22\]](#page-5-0) during intercalation of PbI_2 by some amines: aniline, p-toluidine, hexylamine and p-nitroaniline (Table 4).

Table 3

Changes in vibrational modes of $NH₃$ in the compounds with $PbI₂$ [this work] and in crystal $NH₃$ [\[29\]](#page-6-0)

		v_3 (cm ⁻¹) Δv_3 (cm ⁻¹) v_1 (cm ⁻¹) Δv_1 (cm ⁻¹)		
Gaseous $NH3$	3444		3333	
$PbI_2(NH_3)_{2,47}$	3433	11	3320	13
$PbI_2(NH_3)_4$	3427	17	3304	29
Gaseous $NH3$ [29]	3444		3336	
Crystal $NH3$ [29]	3378	66	3223	113

The anomalous shifts of $v_3(NH)$ and $v_1(NH)$ modes of ammonia in comparison with the shifts for other amines can be caused by the absence of radicals in the ammonia molecule, the radicals imposing steric restrictions on interaction between atom of hydrogen of the N–H bond and electron-donating atom of iodine.

The experimental results obtained can be explained by the existence of the interaction both between hydrogen atom of NH_3 and iodine atom as well as between Pb cation and nitrogen atom during the formation of the compounds PbI_2 with NH₃. That is in good agreement with the data of the works [10,11] where modelling the changes in the band structure of PbI_2 during intercalation resulted in the conclusion that interaction between the iodine $5p_z$ electron and intercalate substantially contributed to the changes in band structure of $PbI₂$.

4. Conclusions

The consequent changes in the crystal structure of $PbI₂$ during incorporating $NH₃$ molecules and the changes in vibrational spectra of incorporated $NH₃$ molecules during changing stoichiometry of the $PbI₂$ $NH₃$ complex were studied. The following four compounds were found: $PbI_2(NH_3)_{1,3}$ (I and II), $PbI_2(NH_3)_{2.47}$ (III) and $PbI_2(NH_3)_4$ (IV). Under the conditions of our experiment three phases $PbI_2(NH_3)_{1,3}$ (I and II) and $PbI_2(NH_3)_{2.47}$ were intermediated phases, and the phase $PbI_2(NH_3)_4$ in ammonia atmosphere at a pressure of 760 Torr and at room temperature was a thermodynamic stable phase. The formation of the complex compound $PbI_2(NH_3)_4$ passed through two stages. Two compounds $PbI_2(NH_3)_{1,3}$ with the lattice constants $a_1 = 4.92 \text{ Å}$, $c_1 = 7.62 \text{ Å}$ and $a_1 = 4.92 \text{ Å}$, $c_2 = 7.49 \text{ Å}$ in $P\bar{3}ml$ space group and the PbI₂(NH₃)_{2.47} compound with the lattice constants $a = 10.238(11)$ Å, $b = 12.061(8)$ Å, $c = 9.587(8)$ Å, $\beta = 107.73(7)^\circ$ in monoclinic system were formed at the first stage of the reaction. At the second stage of the reaction the $PbI_2(NH_3)_4$ compound with the lattice constants $a = 19.894(23)$ Å, $b = 9.345(27)$ Å, $c = 7.021(10)$ Å in orthorhombic system was formed. Anomalous shifts of symmetric $v_1(NH)$ and antisymmetric $v_3(NH)$ modes of the N–H bond of $PbI_2(NH_3)_{2.47}$ and $PbI_2(NH_3)_4$ in comparison with the shifts for other compounds $PbI₂$ with amines were found.

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